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Hierarchical Porous Carbon/MnO₂ Hybrids as Supercapacitor Electrodes

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Hybrid electrodes of hierarchical porous carbon (HPC) and manganese oxide (MnO_2) were synthesized using a fast surface redox reaction of potassium permanganate under facile immersion methods. The HPC/MnO₂ hybrids had a number of micropores and macropores and the MnO_2 nanoparticles acted as a pseudocapacitive material. The synergistic effects of electric double-layer capacitor (EDLC)-induced capacitance and pseudocapacitance brought about a better electrochemical performance of the HPC/MnO₂ hybrid electrodes compared to that obtained with a single component. The hybrids showed a specific capacitance of 228 F g⁻¹ and good cycle stability over 1000 cycles.

Keywords: Porous Carbons, Manganese Oxide, Hybrids, Supercapacitor.

1. INTRODUCTION

Supercapacitors have attracted increasing attention because they have higher power density and longer cycle ability than rechargeable batteries.1 Supercapacitors can be classified according to their energy-storage mechanism into two categories: electric double-layer capacitors (EDLCs) and pseudocapacitors. EDLCs come from the pure electrostatic charge accumulated at the electrode/electrolyte interface. Therefore, EDLCs exhibit good cycle stability and high power density. However, EDLCs require an electrode with a high specific surface area and pores of size of the ions.² Other category is pseudocapacitors, in which fast and reversible faradic processes take place due to an electro-active species. Transition-metal oxides such as ruthenium oxide (RuO_2) or manganese oxide (MnO_2) and electrically conductive polymers have been used as pseudocapacitive electrode materials.³ In particular, MnO₂ has drawn much attention as a pseudocapacitive electrode material for supercapacitors owing to its low cost, natural abundance, environmental compatibility, and high theoretical specific capacitance (~ 1100 F g⁻¹).⁴ However, the poor cycle stability and low electrical conductivity of MnO₂ have limited its practical application as a pseudocapacitive material.5

Recently, carbon-based materials with pseudocapacitive effects have been reported.^{6–8} Carbon-based materials containing electro-active heteroatoms have exhibited great improvement of the specific capacitance owing to both EDLC-induced capacitance and the pseudocapacitive effect. These results suggest that hybrids based on carbonbased materials with a high specific surface area and a metal oxide can be a superior electrode material. In this study, we made macroporous carbon-based material from cellulose.9 The carbon pores are totally open and interconnected, which is advantageous in terms of mass transfer of the reactants.^{10, 11} These porous structures can provide highly efficient electroytle-ion transport through the macropores, micropores and mesopores. The hierarchical porous carbon (HPC), which has a high surface area and a macroporous structure, was selected as a part of the electrode, with pseudocapacitive MnO₂ nanoparticles homogeneously incorporated on the surface of HPC. The HPC/MnO₂ nanoparticle hybrid electrodes exhibited a better electrochemical performance by synergistic effects induced by its EDLC-based and its pseudocapacitive performance.

2. EXPERIMENTAL DETAILS 2.1. Preparation of HPC/MnO₂ Hybrids

HPC was prepared using a previously reported procedure, in which a mixture of 7 wt% NaOH, 12 wt% urea, and 81 wt% water was prepared and pre-cooled to -12 °C for 2 h. Then, 4 wt% cotton cellulose (Aldrich) was immersed in the mixture solution, which was then vigorously stirred

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for approximately 5 min at room temperature. The mixture solution was frozen at -196 °C and then freeze-dried for 3 days. The resulting cryogel was carbonized by heating from room temperature to 700 °C for 2 h at a heating rate of 10 °C/min and an Ar flow rate of 200 mL/min. The solution was then washed using distilled water and ethanol and dried in a vacuum oven at 30 °C for 24 h. The carbon material was ground and added into 200 mL of a 2 mM KMnO₄ solution under continuous stirring at 70 °C for 5 min. After the reaction was completed, the sample was washed using distilled water and ethanol to eliminate unreacted MnO₄ and dried in an oven at 60 °C.

2.2. Characterization

The morphology of the hybrid samples was observed using transmission electron microscopy (TEM, CM200, Philips). X-ray diffraction (XRD, Rigaku DMAX-2500) analysis of the samples was performed using Cu K α radiation (wavelength $\lambda = 0.154$ nm) operated at 40 kV and 100 mA. X-ray photoelectron spectroscopy (XPS, AXIS-HIS, Kratos Analytical, Japan) was performed using monochromated Mg K α radiation (hv = 1500 eV). A twoelectrode cell configuration was used to measure the performance of the samples as electrodes for supercapacitors. For the aqueous system, 5 wt% polytetrafluoroethylene (PTFE, Sigma-Aldrich, 60 wt% dispersion in H_2O) was added to the samples as binder. Typically, the samples and PTFE were mixed to form a paste using a mortar and pestle, rolled into sheets of uniform thickness whit a thickness range from 40 to 50 μ m (from sheet to sheet), and punched into 1×1 cm diameter electrodes. A typical pair of electrodes had a weight between



Figure 1. TEM images of (a) HPC and HPC/MnO₂ hybrids with reaction times of (b) 30 s, (c) 5 min, and (d) 10 min (the arrows are a state in which the manganese oxide is aggregated).

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Figure 2. Schematic diagram of HPC/MnO₂ hybrids fabricated using different reaction times.

2.5 and 3.0 mg after drying overnight at 100 °C. 0.5 M Na_2SO_4 (Sigma-Aldrich, 99%) was used as aqueous electrolyte. The electrodes and a porous polypropylene separator (Whatman GF/B) were sandwiched together in a



Figure 3. (a) Nitrogen adsorption and desorption isotherm curves and (b) pore size distribution of the samples (black squares: HPC, open circles: HPC/MnO₂ hybrids).



Figure 4. (a) XRD patterns of MnO_2 and HPC/MnO_2 hybrids. (b) XPS survey spectra and (c) the magnified view of the Mn 2p region of the hybrids.

stainless steel cell to accomplish the fully assembled twoelectrode system. The electrochemical performances of the samples were evaluated by cyclic voltammetry (CV) and chronopotentiometry.

3. RESULTS AND DISCUSSION

The TEM image of Figure 1(a) exhibits the threedimensional (3-D) macroporous structure of HPC. The morphology of the HPC/MnO₂ hybrids at different magnifications is shown in Figures 1(b) and (c). It can be seen that needle-like MnO2 is distributed on the carbon surface and inner pores. Despite the fact that KMnO₄ is a strong oxidizing agent, the carbon morphology was well preserved. Experimenting with different reaction times and quantities of MnO₂ [Figs. 1(b)-(d)], optimal HPC/MnO₂ hybrids were prepared. With a reaction time of 5 min and 2 mM of KMnO₄ at 70 °C, MnO₂ was homogeneously introduced onto the surface of HPC [Fig. 1(c)]. Figure 2 shows a schematic image of the HPC/MnO₂ hybrids fabricated with different reaction times. The pore characteristics of HPC and HPC/MnO₂ are shown in Figures 3(a), (b). Nitrogen adsorption and desorption isotherms of HPC and HPC/MnO₂ showed IUPAC type-I and IV hybrid shapes. The surface area of HPC is 957.1 m² g⁻¹, which means that the surface areas of micropores (613.1 m² g⁻¹) are twice as large as that of mesopores (343.9 m² g⁻¹). In the case of HPC/MnO₂, the surface area was 651 m² g⁻¹, which means that the surface areas of micropores (410.9 $\text{m}^2 \text{g}^{-1}$) and mesopores $(240.1 \text{ m}^2 \text{ g}^{-1})$ make up. Although the surface area of HPC/MnO₂ was reduced by MnO₂ on carbon, synergistic effects of HPC and MnO2 were observed by maintaining the pore structure, which incresed the capacitance of the HPC/MnO₂ hybrids in comparison to sole HPC.¹² Figure 4(a) shows typical XRD patterns of the as-prepared MnO₂ and HPC/MnO₂ hybrids. The presence of HPC and



Figure 5. Cyclic voltammograms of HPC and HPC/MnO₂ hybrids at a scan rate of 10 mV s⁻¹. (b) Galvanostatic charge/discharge curve of the HPC/MnO₂ hybrids at a current density of 1 A g⁻¹ in a potential window of 0 to 1.0 V. (c) Specific capacitances of HPC and HPC/MnO₂ hybrid electrodes as a function of scan rates. (d) Cyclic stability of the HPC/MnO₂ hybrid electrodes at a scan rate of 50 mV s⁻¹ over 1000 cycles.

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MnO₂ in a mixed crystalline or amorphous structure was confirmed from the XRD patterns.¹⁴ The two broad peaks at around 12° and 37° can be indexed according to MnO₂. The carbon peak appeared at around 26°, with a second broad, weak peak at around 44°, which is characteristic of graphitic carbon. In this sample, the three peaks around 12°, 26°, and 37° were assigned to the crystal planes (110), (002), and (211) of α -MnO₂, respectively.¹³ The XRD pattern of the HPC/MnO₂ hybrids shows an amorphous structure. The chemical configurations of the HPC/MnO₂ hybrids were characterized by XPS. The peaks of Mn $(2p_{3/2}, 2p_{1/2})$, O 1s, and C 1s are shown in the survey spectrum [Fig. 4(b)]. The Mn $2p_{3/2}$ peak is centered at 642.1 eV and the Mn $2p_{1/2}$ peak at 653.8 eV, which is in good agreement with the reported data for Mn $2p_{3/2}$ and $2p_{1/2}$ in MnO₂ [Fig. 4(c)].¹⁵ The XPS data coincide with the XRD data.

The evaluation of the electrochemical performance of the HPC/MnO₂ hybrids was conducted in a 0.5 M Na₂SO₄ aqueous electrolyte. Figure 5(a) shows the CV curves of the HPC and HPC/MnO₂ hybrids over a potential range from 0 to 1 V at a scan rate of 10 mV s⁻¹. It is noticed that the curves are somewhat rectangular, which indicates almost ideal capacitance behavior. The specific capacitances of the HPC and HPC/MnO₂ hybrids were 161 and 228 F g^{-1} , respectively. The hybrid electrodes showed a capacitance that was approximately 41.6% higher than that of HPC. The galvanostatic charge/discharge curves of the HPC and HPC/MnO₂ hybrids are shown in Figure 5(b). It shows that the charging curves are symmetric with their discharging counterparts. The specific capacitances of the HPC and HPC/MnO₂ hybrids were also caculated from the galvanostatic charge-discharge curves at a discharging rate of 1 A g^{-1} and reached 110 and 180 F g^{-1} , respectively. The variation in the specific capacitance of the HMC/MnO₂ hybrids as a function of the scan rates is plotted in Figure 5(c). The specific capacitance decreased on increasing scan rates. However, the specific capacitance of the HPC/MnO₂ hybrids (103 F g⁻¹) was maintained at a relatively high scan rate of 200 mV s⁻¹. Also, across the entire range of scan rates, the HPC/MnO₂ hybrids showed higher capacitnaces than HPC. This is a result of the synergistic effects of EDLC-based HPC and the pseudocapacitive effects of MnO₂. Moreover, the HPC/MnO₂ hybrids can increase the effective contact area between MnO₂ and the electrolyte, which results in high electrochemical utilization of MnO₂. For the HPC/MnO₂ hybrids, cycling tests were carried out for 1000 cycles [Fig. 5(d)]. After 1000 cycles at a scan rate 50 mV s^{-1} , the capacitances of the HPC/MnO₂ hybrids decreased to 11% of the initial capacitance, which was attributed to HPC on covering the MnO₂ by reducing the stress and proper pore structure.

4. CONCLUSIONS

HPC/MnO₂ hybrids were prepared by using cellulosebased materials and a fast surface redox reaction of potassium permanganate under facile immersion methods. HPC/MnO₂ hybrids were prepared with 2 mM KMnO₄ and a reaction time of 5 min at 70 °C. The HPC/MnO₂ hybrids showed a specific capacitance of 228 F g⁻¹ at a scan rate of 10 mV s⁻¹, and a specific capacitance of 103 F g⁻¹ was maintained at a relatively high scan rate of 200 mV s⁻¹. Furthermore, a stable electrochemical performance was maintained for 1000 cycles.

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References and Notes

- **31.** P. Simon and Y. Gogotsi, Nat. Mater. 7, 845 (2008).
- V 2. Yo St.Yun, S. Y. Cho, J. Shim, B. H. Kim, S.-J. Chang, S. J. Baek, Y. S. Huh, Y. Tak, Y. W. Park, S. Park, and H.-J. Jin, *Adv. Mater*. 25, 1993 (2013).
 - 3. V. Khomenko, E. Raymundo-Pinero, and F. Beguin, J. Power Sources 153, 183 (2006).
 - G. An, P. Yu, M. Xiao, Z. Liu, Z. Miao, K. Ding, and L. Mao, Nanotechnology 19, 275709 (2008).
 - 5. Z. Li, Y. Mi, X. Liu, S. Liu, S. Yang, and J. Wang, J. Mater. Chem. 21, 14706 (2011).
 - 6. D. Hulicova-Jurcakova, M. Seredych, G. Q. Lu, and T. J. Bandosz, *Adv. Funct. Mater.* 19, 438 (2009).
 - 7. Y. S. Yun, H. H. Park, and H.-J. Joon, Materials 5, 1258 (2012).
 - 8. M. Evans, E. Halliop, and J. Macdonald, Carbon 37, 269 (1999).
 - Y. S. Yun, J. Y. Shim, Y. S. Tak, and H.-J. Jin, *Rse Adv.* 2, 4353 (2012).
- M. Lillo-Rodenas, D. Lozano-Castello, D. Cazorla-Amoros, and A. Linares-Solano, *Carbon* 39, 751 (2001).
- S. B. Ma, K. Y. Ahn, E. S. Lee, K. H. Oh, and K. B. Kim, *Carbon* 45, 375 (2007).
- 12. Y. S. Yun, C. B. Im, H. H. Park, I. G. Hwang, Y. S. Tak, and H.-J. Jin, J. Power Sources 234, 285 (2013).
- Y. U. Jeong and A. Manthriram, J. Electrochem. Soc. 149, A1419 (2002).
- 14. J. Yan, Z. Fan, T. Wei, J. Cheng, B. Shao, K. Wang, L. Song, and M. Zhang, J. Power Sources 194, 1202 (2009).
- S. W. Lee, J. H. Kim, S. Chen, P. T. Hammond, and Y. Shao-Horn, ACS Nano 4, 3889 (2010).

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